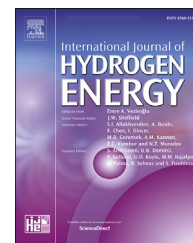


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Short Communication

Boosting visible light driven hydrogen production: Bifunctional interface of Ni(OH)₂/Pt cocatalyst on TiO₂

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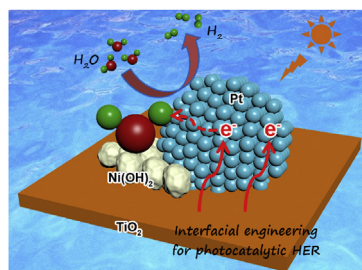
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HIGHLIGHTS

- Bifunctional Ni(OH)₂/Pt interface was successfully constructed on TiO₂ nanotubes.
- Ni(OH)₂/Pt interface facilitates the adsorption of H₂O.
- Ni(OH)₂/Pt interface promotes the separation of electron/hole pairs.
- A high visible-light driven hydrogen evolution activity of 5875 μmol g⁻¹ h⁻¹ was achieved for Ni(OH)₂/Pt/TiO₂ sample.

GRAPHICAL ABSTRACT



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ABSTRACT

Sunlight driven hydrogen production by water splitting represents a sustainable approach for hydrogen energy utilization, however, solo semiconductor could not meet the current demand. Herein, as proof of concept, we put forward a bifunctional interface of Ni(OH)₂/Pt cocatalyst loaded on anatase TiO₂ through a selective deposition process. Ni(OH)₂/Pt interface facilitates H₂O adsorption by providing two adsorption sites: O is selectively adsorbed on Ni(OH)₂ and H adsorbed on metallic Pt. Beyond that, Ni(OH)₂/Pt interface also extends light absorption range of TiO₂ to visible light region and accelerates the separation of photo-generated electrons/holes for semiconductor TiO₂. Benefiting from this interfacial engineering in semiconductor, the visible-light driven hydrogen production performance

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Visible-light driven
Hydrogen production
Ni(OH)₂

of the as-obtained Ni(OH)₂/Pt/TiO₂ sample is greatly boosted with a rate of 5875 $\mu\text{mol g}^{-1} \text{h}^{-1}$, 2.6 times higher than that of Pt/TiO₂.

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Introduction

Solar energy utilization represents a sustainable approach that has a bright future to solve the current energy crisis [1–5]. Hydrogen production from water splitting under sunlight with semiconductor catalysts seems to be the most possible means of implementation [6,7]. Typically, semiconductors usually absorb suitable light to produce electrons/holes, and the active electrons are migrated to the surface of semiconductors to participate reduction reaction (i.e., hydrogen production) [8–10]. However, photocatalytic hydrogen production still suffers from several bottlenecks: i) Weak visible-light absorption ability which is determined by the wide band gap of semiconductors. Such as anatase TiO₂, its band gap is up to 3.2 eV and usually only absorbs UV-light, which only accounts for 5% of total energy in the sunlight [11,12]; ii) Less photo-generated reactive electrons, which often recombine with holes before migrating to the catalyst surface and participating reactions [13]; iii) Insufficient catalytic sites to adsorb and convert water molecule into hydrogen [14,15].

The above bottlenecks for sunlight driven hydrogen production can be greatly alleviated by hybridization of semiconductors with noble metal nanocrystals as cocatalysts. A Schottky junction is naturally formed in metal/semiconductor interface, which facilitates the separation of electrons/holes and provides active sites for hydrogen evolution [16,17]. Pt is seemed as the most active single metallic cocatalyst for hydrogen evolution in acidic medium. H⁺ has very low adsorption energy on Pt surface and 2H_{ad} are combined to hydrogen via a Tafel step with fast kinetics [18,19]. However, in neutral and alkaline media, water splitting is triggered by adsorption of H₂O and subsequent formation of a reactive hydrogen intermediate H_{ad} in a process of Volmer step (H₂O + Pt + e⁻ → Pt-H_{ad} + OH⁻). 2H_{ad} are coupled to complete hydrogen evolution reaction followed by Heyrovsky step (H₂O + e⁻ + Pt-H_{ad} → H₂ + OH⁻ + Pt) or Tafel step (2Pt-H_{ad} → H₂ + 2 Pt) [20]. Stemmed from the above steps with slow kinetics, hydrogen evolution activity is greatly declined in neutral and alkaline media. Therefore, photocatalytic water splitting in neutral and alkaline media always displays much lower activity with solo Pt cocatalysts at the mercy of adsorbed water [21,22]. Recent discoveries in electrochemical water splitting reaction indicate that the existence of Ni(OH)₂/Pt interface can boost hydrogen evolution activity in alkaline media by providing two kinds of adsorption sites for O and H atoms, respectively [23,24]. Ni(OH)₂ as the O adsorption site promotes the splitting of HO–H and the 2H_{ad} adsorbed on Pt surface are coupled via Tafel path [25]. Therefore, hydrogen evolution ability is boosted as Tafel process possesses fast reaction kinetics.

Enlightened by the above discoveries, Ni(OH)₂/Pt interface seems to be an ideal direction toward efficient hydrogen production in neutral solution. Herein, as a proof of concept, we construct a bifunctional interface of Ni(OH)₂/Pt cocatalyst by coating discrete amorphous Ni(OH)₂ layer on Pt loaded TiO₂ nanotube. Different from the conventional Pt cluster decorated Ni(OH)₂ nanoparticle cocatalyst (Pt/Ni(OH)₂/TiO₂ sample) [26], our catalyst design possess the following unique structure merits: i) the direct contact of TiO₂ and Pt (Pt/TiO₂ interface) naturally forms the Schottky junction which can extend light absorption range for TiO₂ to visible light region and have more ability to intensify the separation of photo-generated electron/holes than Ni(OH)₂/TiO₂ interface. ii) The electron accumulation on Pt site shortens the electron transfer path than Pt/Ni(OH)₂/TiO₂ sample as the photo-generated electrons are transformed to Ni(OH)₂ and then to Pt. iii) Ni(OH)₂/Pt interface provide two kinds of adsorption sites to bind O on Ni(OH)₂ and H on Pt sites, respectively. Benefitting from the above merits, the visible-light driven hydrogen production performance for Ni(OH)₂/Pt/TiO₂ sample is greatly enhanced in reference with that of Pt/TiO₂, Ni(OH)₂/TiO₂ and pristine TiO₂ nanotubes. This work provides a promising approach to construct bifunctional cocatalyst by the combination of interface engineering and Schottky junction.

Experimental

Synthesis of Ni(OH)₂/Pt/TiO₂ hybrid nanotubes

High-quality of pristine TiO₂ nanotubes were synthesized according to the previous literature [27]. In order to deposit Pt nanoparticles on TiO₂ nanotubes, the as-obtained TiO₂ (0.5 g) nanotubes were dispersed in H₂PtCl₆ (2.5 mM) aqueous solution, then excess NaBH₄ (10.25 mL, 50 mM) aqueous solution were added into the mixture under continuous stir, which ensure the complete reduction of PtCl₆²⁻. The product was separated through centrifugation, washed with water and ethanol three times, respectively. To further cover ultrathin amorphous Ni(OH)₂ shell on Pt/TiO₂ nanotubes, 0.5 g Pt/TiO₂ were further dispersed in 50 mL NaOH (0.5 M) and stirred for 2 h, 27 mL Ni(NO₃)₂ (10 mM) aqueous solution was added into the mixture drop by drop. After 2 h reaction, the final sample was collected by centrifugation and washed with pure water three times, and dried in vacuum oven at 60 °C. The synthetic protocol for Ni(OH)₂/TiO₂ nanotubes was similar with that of Ni(OH)₂/Pt/TiO₂ except with 0.5 g TiO₂ nanotubes.

Photocatalytic hydrogen evolution

The photocatalytic hydrogen evolution reaction from water was performed using a commercial online photocatalytic

hydrogen production system (AuLight, Beijing, CEL-SPH2N). Specifically, 0.1 g of $\text{Ni(OH)}_2/\text{Pt}/\text{TiO}_2$ nanotubes were added into a 100 mL aqueous solution containing 80 mL water, 20 mL methanol in a quartz reactor. The openings for the whole system were sealed with septums made of silicone rubber at ambient temperature and pressure. Prior to the reaction, the mixture was evacuated into vacuum and pumped by continuous Ar gas flow several times with stirring to remove O_2 and CO_2 dissolved in the solution. A 300 W xenon arc lamp equipped with 400 nm cut optical filter was used as a light source to trigger the photocatalytic reaction. The actual light intensity in the center of quartz reactor was measured by a visible light radiometer (made in the photoelectric instrument factory of Beijing Normal University, China). The focused light intensity on the liquid surface was maintained to 100 mW cm^{-2} . Gas product (H_2) was analyzed using an online gas chromatograph (SP7800, TCD, molecular sieve 0.5 nm, N_2 carrier, Beijing Keruida Limited) and the actual produced H_2 amount was quantified using an internal standard method.

Results and discussion

In order to investigate the impact of $\text{Ni(OH)}_2/\text{Pt}$ interface on photocatalytic hydrogen evolution activity, high-quality one dimensional anatase TiO_2 nanotubes with curled layered structures were employed as semiconductor substrates, as shown in Fig. S1. Then, Pt nanoparticles were loaded on TiO_2 nanotubes by a careful reduction of H_2PtCl_6 with NaBH_4 , as shown in Fig. S2. To further implement the deposition of ultrathin discrete Ni(OH)_2 shell, Pt/TiO_2 nanotubes were infiltrated in NaOH aqueous solution and then $\text{Ni(NO}_3)_2$ solution was dropwise added. This accurate synthetic protocol ensures the formation of ultrathin Ni(OH)_2 shell on Pt/TiO_2 nanotubes and partial exposure of Pt nanoparticles for hydrogen evolution reaction. Fig. 1a shows the powder X-ray diffraction (XRD) patterns of $\text{Ni(OH)}_2/\text{Pt}/\text{TiO}_2$ sample in reference with pristine TiO_2 , Pt/TiO_2 and $\text{Ni(OH)}_2/\text{TiO}_2$ nanotubes. All primary diffraction peaks are well matched with standard anatase TiO_2 (JCPDS No. 21–1272) for all samples. Slightly raised peaks located at 39.8° for $\text{Ni(OH)}_2/\text{Pt}/\text{TiO}_2$ and Pt/TiO_2 samples are attributed to (111) plane of face-centered cubic Pt (JCPDS No. 87–0640). There are no additional peaks could be assigned to Ni(OH)_2 , maybe due to its amorphous nature and/or low content. The field emission scanning electron microscope (SEM) image in Fig. 1b confirms the excellent one-dimensional structure maintained for $\text{Ni(OH)}_2/\text{Pt}/\text{TiO}_2$ sample. Low resolution transmission electron microscopy (TEM) image in Fig. 1c verifies the existence of Pt nanoparticles and one-dimensional tubular structure.

High resolution TEM (HRTEM) image (Fig. 1d) displays the continuous lattice fringes with a distance of 2.0 Å which can be attributed to Pt (200) plane. Also, lattice fringes with a distance of 3.5 Å (Fig. 1e) are ascribed to (101) plane of anatase TiO_2 . Different from the smooth outermost surface for Pt/TiO_2 and pristine TiO_2 nanotubes (Figs. S1 and S2), $\text{Ni(OH)}_2/\text{Pt}/\text{TiO}_2$ nanotubes display rough surfaces, indicating that ultrathin Ni(OH)_2 layer with about a thickness of 1.0 nm was successfully coated on the outside after a careful examination in locally amplified HRTEM image (Fig. 1f). There is no obvious

lattice fringes for the rough region, indicating that the Ni(OH)_2 layer is amorphous. To further verify the crystal nature, we taken out Pt/TiO_2 nanotubes from the standard synthetic protocol, the products are identified to be Ni(OH)_2 by XRD pattern (Fig. S3), suggesting that the outmost amorphous layer for $\text{Ni(OH)}_2/\text{Pt}/\text{TiO}_2$ is Ni(OH)_2 . It should be noted that Pt nanoparticles are not completely covered by amorphous Ni(OH)_2 layer (Fig. 1d–f), guaranteeing the existence of abundant $\text{Ni(OH)}_2/\text{Pt}$ interface. The partial exposure of Pt nanoparticles may be due to the high electrons' concentration on Pt, which leads to its electronegativity and the electrostatic repulsion to Ni(OH)_2 [28]. To further investigate the Ni(OH)_2 content for $\text{Ni(OH)}_2/\text{Pt}/\text{TiO}_2$ hybrid sample, thermogravimetric (TG) analysis was performed, as shown in Fig. S4. The hybrid sample starts to decompose at 220°C , which can be assigned to the conversion of Ni(OH)_2 to NiO , consistent with the reported result for Ni(OH)_2 [29]. And 1.8 wt% weight loss between 220 and 350°C was achieved, suggesting 9.2 wt% of Ni(OH)_2 for $\text{Ni(OH)}_2/\text{Pt}/\text{TiO}_2$ sample. Ni(OH)_2 content was also determined by inductively coupled plasma atomic emission spectrometry (ICP-AES) which reveals 8.5 wt% of Ni(OH)_2 , consistent with TG data.

In order to further investigate the species and chemical state of the elements for the sample, X-ray photoelectron spectroscopy (XPS) measurement was employed. The survey spectrum in Fig. 2a confirms the existence of the elements of Ti, O, Ni and Pt for $\text{Ni(OH)}_2/\text{Pt}/\text{TiO}_2$ sample. The high resolution spectrum (Fig. 2b) reveals the peaks located at 856.1 and 873.6 eV are ascribed to $2p_{3/2}$ and $2p_{1/2}$ of Ni^{2+} , respectively [30]. The binding energy for Ni $2p_{3/2}$ located at 856.1 eV is the characteristic peak for Ni(OH)_2 with associated intensive satellite peaks at 862.0 eV and 880.0 eV, further confirming the outermost layer of sample is Ni(OH)_2 , consistent with XRD and TG results [31,32]. Fig. 2c displays two binding energies at 71.0 and 74.3 eV, which are attributed to be $4f_{7/2}$ and $4f_{5/2}$ of typical Pt^0 , respectively. Binding energies located at 458.6 and 464.5 eV (Fig. 2d) are ascribed to $2p_{3/2}$ and $2p_{1/2}$ of Ti^{4+} , respectively. The XPS data of Ti^{4+} and Ni^{2+} manifests that the formation of Pt/TiO_2 , $\text{Ni(OH)}_2/\text{Pt}$ and $\text{Ni(OH)}_2/\text{TiO}_2$ interfaces will not affect the chemical state of pristine TiO_2 and Ni(OH)_2 . We also tested specific surface areas of all samples using N_2 adsorption-desorption measurements, $\text{Ni(OH)}_2/\text{Pt}/\text{TiO}_2$ nanotubes exhibit a comparative specific surface area of $146.2 \text{ m}^2 \text{ g}^{-1}$ with that of Pt/TiO_2 ($145.7 \text{ m}^2 \text{ g}^{-1}$), $\text{Ni(OH)}_2/\text{TiO}_2$ ($144.0 \text{ m}^2 \text{ g}^{-1}$) and TiO_2 samples ($140.1 \text{ m}^2 \text{ g}^{-1}$) (Fig. S5 and Table S1). The cladding of ultrathin Ni(OH)_2 layer doesn't shield the whole surface of TiO_2 (Fig. 1e and f). The large specific surface area for $\text{Ni(OH)}_2/\text{Pt}/\text{TiO}_2$ sample inherited from ultrathin TiO_2 nanotubes means that more active catalytic sites are provided for hydrogen evolution reaction.

The formation of $\text{Ni(OH)}_2/\text{Pt}$ interface provides an ideal platform to investigate the impact of interfacial interaction for Pt cocatalyst on photocatalytic hydrogen production. All photocatalytic hydrogen evolution activities are assessed under simulative visible-light irradiation ($\lambda > 400 \text{ nm}$) using commercialized photocatalytic reaction system with the same catalyst dosage of 0.1 g and methanol as sacrificial agent. In order to make a better comparison, we have also made $\text{Ni(OH)}_2/\text{TiO}_2$ nanotubes as reference catalyst (Fig. S6, Table S1). Fig. 3a shows the curves of hydrogen evolution amount

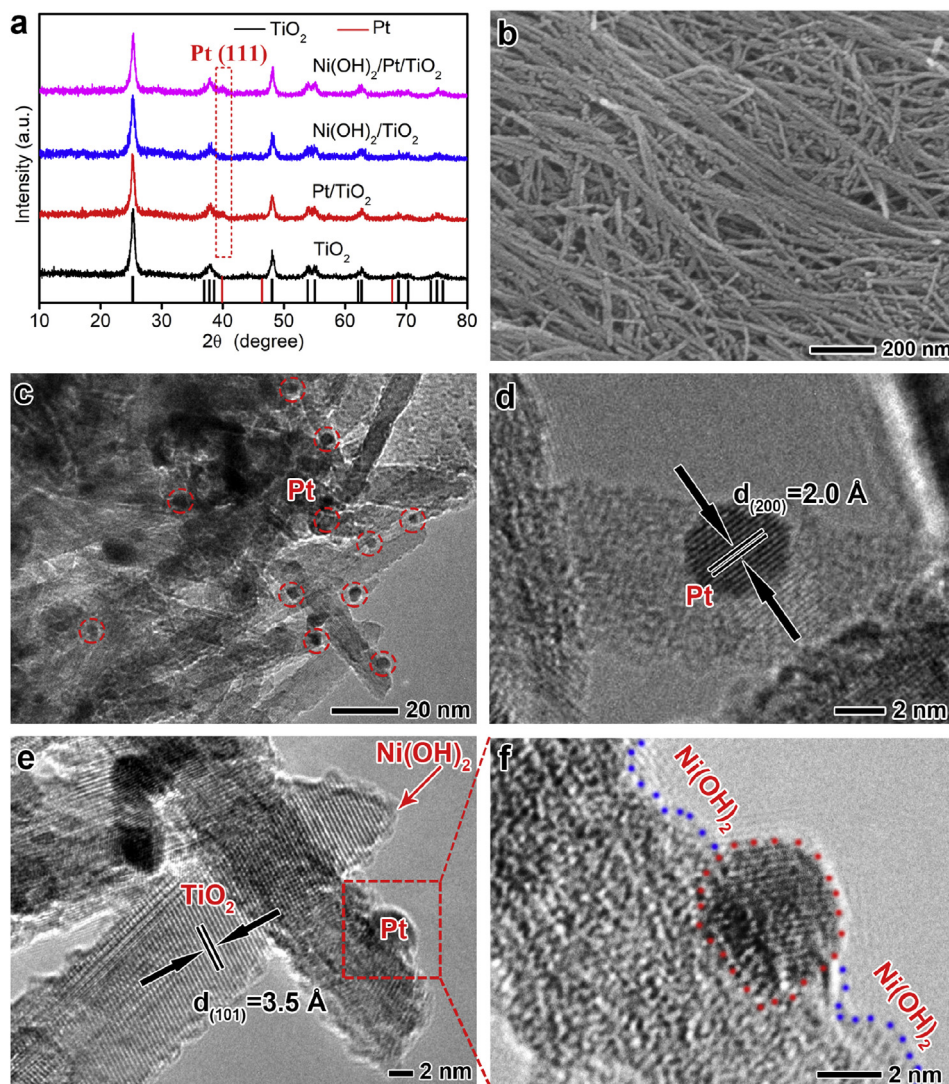


Fig. 1 – (a) Powder X-ray diffraction patterns, (b) scanning electron micrograph, (c) transmission electron photograph, and (d–f) high-resolution transmission electron photographs of Ni(OH)₂/Pt/TiO₂ hybrid nanotubes.

versus illumination time. For all samples, a linear relationship between hydrogen evolution amount and illumination time indicates that hydrogen is really originated from photocatalytic reaction. Specifically, Ni(OH)₂/Pt/TiO₂ sample achieves a hydrogen production rate of 5875 $\mu\text{mol g}^{-1} \text{h}^{-1}$, which is 2.6, 6.4 and 43.8 times higher than that of Pt/TiO₂ (2241 $\mu\text{mol g}^{-1} \text{h}^{-1}$), Ni(OH)₂/TiO₂ (912 $\mu\text{mol g}^{-1} \text{h}^{-1}$) and pristine TiO₂ (134 $\mu\text{mol g}^{-1} \text{h}^{-1}$) samples, respectively. The greatly enhanced photocatalytic hydrogen production manifests that the interfacial engineering by the formation of Ni(OH)₂/Pt interface really promotes the hydrogen evolution reaction in neutral solution.

Apart from Ni(OH)₂/Pt interface, the existence of Ni(OH)₂/TiO₂ interfacial interaction also contributes to the enhanced hydrogen evolution activity, which is verified by the comparison of different activities of Ni(OH)₂/TiO₂ and TiO₂ samples. We also assessed the cycling stability of Ni(OH)₂/Pt/TiO₂ catalyst. Our catalyst produces comparable hydrogen amount

during the five cycles (25 h), indicating the excellent stability for Ni(OH)₂/Pt/TiO₂ nanotubes (Fig. S7).

Upon assessing the photocatalytic hydrogen production activities, it is found that the synergistic effect of Ni(OH)₂ and Pt really contributes to the enhanced photocatalytic hydrogen evolution performance. It is well known that Ni²⁺ in Ni(OH)₂ interacts with HO–H via Ni²⁺–O^{δ-} electrostatic interaction, while H atom binds well with Pt [26,33]. Therefore, Ni(OH)₂/Pt interface strengthens the adsorption of H₂O and promotes its following splitting. And the resulted H_{ad} is recombined to H₂ with the help of Pt catalyst [34]. To verify the synergistic photocatalytic effect in this Ni(OH)₂/Pt/TiO₂ system, FT-IR measurement was performed (Figs. 4a and S8). Fig. 4a shows the FT-IR spectra after infinitesimal H₂O adsorption by subtracting the background spectra of samples without H₂O adsorption. The peak located at around 1630 cm⁻¹ is the bending vibration mode for adsorbed water molecule [35]. Ni(OH)₂/Pt/TiO₂ nanotube possesses a higher bending mode (1638 cm⁻¹) relative to that of Pt/TiO₂ (1627 cm⁻¹) and Ni(OH)₂/

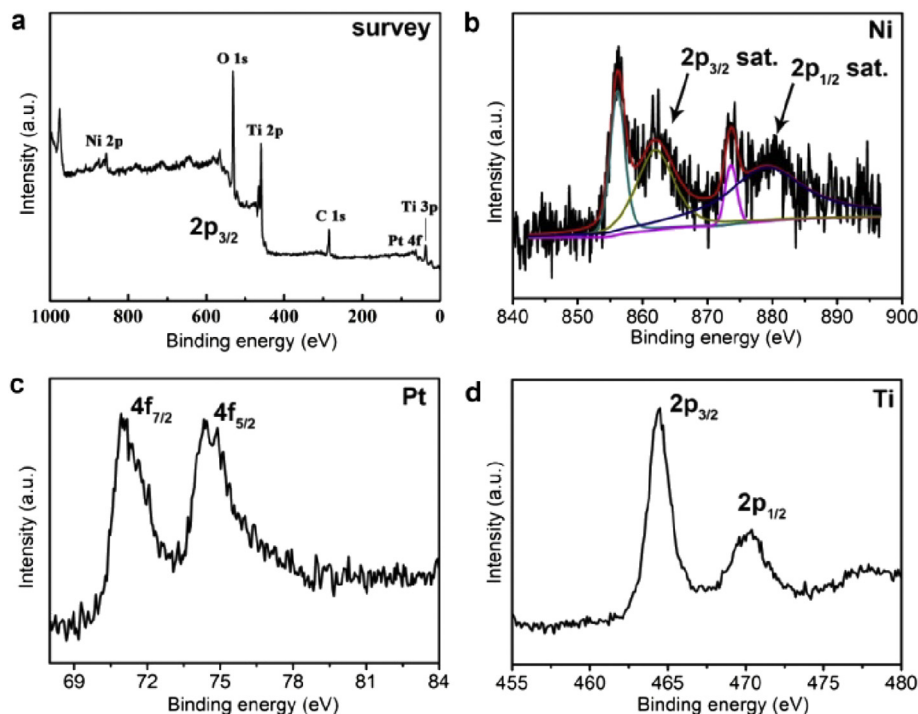


Fig. 2 – X-ray photoelectron spectra of $\text{Ni(OH)}_2/\text{Pt/TiO}_2$ hybrid nanotubes. (a) Survey, (b) Ni 2p, (c) Pt 4f, (d) Ti 2p.

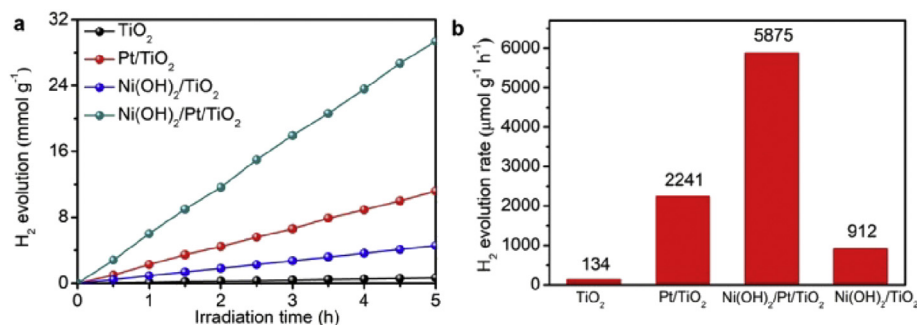


Fig. 3 – (a) The comparison of visible light driven hydrogen production using $\text{Ni(OH)}_2/\text{Pt/TiO}_2$ hybrid nanotubes as catalysts and methanol as sacrificial agent under the irradiation of visible light ($\lambda > 400$ nm) in reference with Pt/TiO_2 , $\text{Ni(OH)}_2/\text{TiO}_2$ and TiO_2 samples.

TiO_2 (1625 cm^{-1}), confirming the synergistic mode of O and H atoms that are fixated on $\text{Ni(OH)}_2/\text{Pt}$ interface, respectively [36]. The promotion effect of the interfacial interaction is also evidenced by Tafel slopes derived from linear sweep voltammetry (LSV) curves recorded in neutral media (Fig. S9). $\text{Ni(OH)}_2/\text{Pt/TiO}_2$ possesses a Tafel slope of 216 mV dec^{-1} , much lower than that of Pt/TiO_2 sample (299 mV dec^{-1}), suggesting a faster water splitting kinetics [37,38].

Apart from the promotion effect on the enhanced water splitting kinetics, $\text{Ni(OH)}_2/\text{Pt}$ interface also contributes to the visible light absorption. We first assessed the light absorption ability of $\text{Ni(OH)}_2/\text{Pt/TiO}_2$ catalyst using ultraviolet diffuse reflection spectra in reference with Pt/TiO_2 , $\text{Ni(OH)}_2/\text{TiO}_2$ and pristine TiO_2 samples (Fig. 4c). The deposition of Pt nanoparticles on TiO_2 extends the light absorption range to visible light region, that is because the valence band electrons of TiO_2

are excited to localized energy levels between Pt/TiO_2 interface at wavelength longer than 400 nm [39,40]. $\text{Ni(OH)}_2/\text{Pt/TiO}_2$ sample exhibits enhanced visible light ($400\text{--}800\text{ nm}$) absorption ability compared to Pt/TiO_2 nanotubes. The UV-vis absorption spectra for TiO_2 and $\text{Ni(OH)}_2/\text{TiO}_2$ samples are almost coincided with each other, indicating that the $\text{Ni(OH)}_2/\text{TiO}_2$ interface will not greatly impact the visible light absorption ability for TiO_2 . Therefore, the enhanced visible light absorption for $\text{Ni(OH)}_2/\text{Pt/TiO}_2$ sample should be ascribed to the formation of $\text{Ni(OH)}_2/\text{Pt}$ interface. The large electronic affinity of Ni(OH)_2 may accelerate the photo-generated electrons to the localized energy level induced by Pt/TiO_2 interface, then visible light absorption ability is enhanced [41].

The band gap for all samples are derived from the transition of Kubelka-Munk function, and $\text{Ni(OH)}_2/\text{Pt/TiO}_2$ possesses a narrowed band gap of 2.72 eV compared with that of pristine

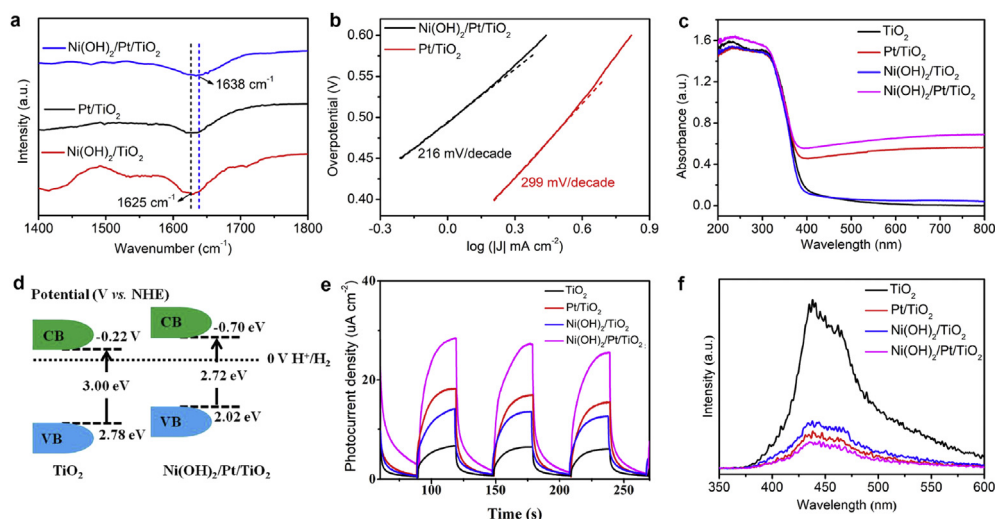


Fig. 4 – (a) Fourier transform infrared spectra of Ni(OH)₂/Pt/TiO₂ and Pt/TiO₂ nanotubes after water adsorption. (b) Tafel plots derived from LSV curves recorded in 0.5 M K₂SO₄ solution using Ni(OH)₂/Pt/TiO₂ and Pt/TiO₂ nanotubes as catalysts. (c) UV–vis absorption spectra. (d) The electronic band structures of TiO₂ nanotube and Ni(OH)₂/Pt/TiO₂ hybrid nanotube. (e) Photocurrent responses versus time (I-t) curves of pristine TiO₂, Pt/TiO₂, Ni(OH)₂/TiO₂ and Ni(OH)₂/Pt/TiO₂ nanotubes under visible light illumination ($\lambda > 400$ nm) at a bias potential of -0.3 V vs. RHE. (f) Photoluminescence spectra using a 320 nm light as excitation light source.

TiO₂ (3.00 eV). It should be noted that TiO₂ nanotubes display a band gap of 3.00 eV, lower than the intrinsic one (3.20 eV) for TiO₂. That is because our nanotubes possess bits of oxygen defects (Fig. S11), which also explain the visible-light hydrogen evolution ability for TiO₂ nanotubes [42,43]. The narrowed band gap indicates that light absorption cut off is further expended to visible light region. To gain the exact valence band, ultraviolet photoelectron spectroscopy (UPS) measurements are employed and the calculated values are 2.78 and 2.02 eV for TiO₂ and Ni(OH)₂/Pt/TiO₂ samples (Fig. S12), respectively. Therefore, the corresponding conduction bands are inferred and shown in Fig. 4d.

The photo-generated electrons are pumped to the surface of Pt nanoparticles due to the relative larger work function of

Pt than that of TiO₂, knowing as Schottky junction [44,45]. The separation efficiency of photo-generated electron/holes can be assessed by photocurrent tests. As shown in Fig. 4e, Ni(OH)₂/Pt/TiO₂ sample displays a much enhanced photocurrent response than Pt/TiO₂, Ni(OH)₂/TiO₂ and TiO₂ samples. The enhanced photocurrent response should be enabled by the following two factors: i) The formation of Ni(OH)₂/Pt interface combined with a Schottky junction intensifies the visible-light absorption and induces more photo-generated electrons (Fig. S13). ii) The existence of Ni(OH)₂ also promotes the separation of carriers for TiO₂ due to its higher electronic transmission ability, as indicated by EIS analysis (Fig. S14) [46,47]. The hypothesis is further verified by comparison the photocurrents of Ni(OH)₂/TiO₂ and pristine TiO₂ nanostructures. To further inspect the practical separation efficiency of electrons/holes pairs under visible-light irradiation, steady-state photoluminescence (PL) spectroscopy measurements with a same amount of samples using a 320 nm excitation light have been conducted, as shown in Fig. 4f. Ni(OH)₂/Pt/TiO₂ sample possesses the lowest PL intensity in reference with Pt/TiO₂, Ni(OH)₂/TiO₂ and TiO₂ samples. The lower emission peak intensity indicates that the recombination of photo-generated electrons/holes is effectively suppressed, consistent with the calculated carrier densities, photocurrent and EIS data [48]. To facilitate understanding, the detailed mechanism of the formation of Ni(OH)₂/Pt cocatalyst on TiO₂ nanotubes to boost photocatalytic hydrogen evolution reaction is schematically illustrated in Fig. 5.

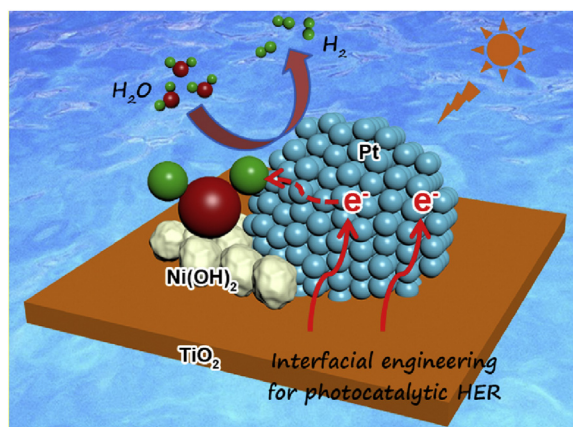


Fig. 5 – Schematic illustration of the mechanism of Ni(OH)₂/Pt cocatalyst on TiO₂ to boost photocatalytic hydrogen evolution reaction.

Conclusions

In summary, we have successfully constructed a Ni(OH)₂/Pt hybrid interface to realize a chemical synergistic mechanism

on promoting water splitting based on their respective interaction to O and H atoms in water. This mechanism is proved by the corresponding FT-IR spectroscopy and electrochemical Tafel measurements. Moreover, Ni(OH)₂/Pt interface further enhances the visible light absorption and elevates the separation efficiency of photo-generated electrons/holes. Benefiting from the above merits, Ni(OH)₂/Pt/TiO₂ hybrid nanotubes display much improved visible light driven photocatalytic hydrogen evolution performance with a rate of 5875 $\mu\text{mol g}^{-1} \text{h}^{-1}$. This work provides insights into design of highly efficient photocatalyst for visible light driven reaction based on interfacial engineering.

Author contributions

The manuscript was written through contributions of all authors. All authors have given approval to the final version of the manuscript.

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Appendix A. Supplementary data

Supplementary data to this article can be found online at <https://doi.org/10.1016/j.ijhydene.2020.04.090>.

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